

# FABRICATED Fe-Mn NANOWIRE AS A BINARY OXIDE BY HYDROTHERMAL TREATMENT

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**ABSTRACT:** Fe-Mn binary oxide with (1:1)Fe:Mn molar ratios was synthesized by a simultaneous oxidation and co-precipitation process. X-Ray Diffraction (XRD) study indicated the formation of nanostructure, strain and dislocations was studied by Williamson Hall. Hydrothermal treatment method give us nanowire structure Fe-Mn (1:1)N.W. morphology of obtained materials was showed by scanning Electron Microscopy (SEM) , imagej program using to compute average particle size average diameter size (30-60)nm for nanoparticles, average length~( 1500-2000)nm and the average diameter 63nm of nanowire. Energy Dispersive X Ray Spectrometry (EDS) spectra showed strong peaks of Fe ,Mn and O for Fe-Mn binary oxide.

**Keywords:** Fe-Mn nanowire, oxidation and co-precipitation, dislocations

## INTRODUCTION:

Metal oxide nanowires such as TiO<sub>2</sub> [1], SiO<sub>2</sub>[2], and Fe<sub>2</sub>O<sub>3</sub> [3,4], have attracted much attention because of their novel applications of nanowires in Magnetic Resonance Imaging, photoluminescence, light sensing and water treatment, so materials science and nanotechnology has expanded rapidly. The inevitable intersection of these three disciplines has set in motion the development of an emerging research area, Magnetic porous Fe–Mn binary oxide nanowires were success-fully fabricated[5]. a variety of synthesis approaches have been developed to produce high quality nanoparticles Generally, nanomaterials should be stable to avoid aggregation [6]. a novel Fe–Mn binary oxide sorbent has been successfully developed, exhibiting high effectiveness in removing. The magnetite can be synthesized by various methods, including: ultrasound irradiation [7], sol-gel [8],thermal decomposition[9] and co-precipitation [10,11]. Thermal decomposition and co-precipitation are most commonly used

## 2. EXPERIMENTAL SECTION

### 2.1. Materials

Chemicals used were of analytical grade. Reaction vessels (glass) were cleaned with 1% HNO<sub>3</sub> and rinsed several times with deionized water before use. iron(II) Sulfate heptahydrate FeSO<sub>4</sub>.7H<sub>2</sub>O,Chaina Potassium KMnO<sub>4</sub> permanganate, panreac spain.

**2.2. Preparation of Fe oxide and Fe–Mn binary oxides** Fe–Mn binary oxides was synthesized at molar ratio(1:1). The Fe–Mn binary oxide with a molar ratio of 1:1was prepared according to a method slightly modified from that proposed by G.S.Zhang *et al.* [6]. A mixture of Potassium permanganate(KMnO<sub>4</sub>) and iron(II) sulfate ) in the ratio 1:1 (0.015) plus heptahydrate (FeSO<sub>4</sub>.7H<sub>2</sub>) were dissolved in 200 mL of deionized water. Under vigorous magnetic stirring, the FeSO<sub>4</sub> solution was added into the KMnO<sub>4</sub> solution simultaneously with 4 M NaOH solution to maintain he pH of the solurion in the range 7 and 8. The suspension was then stirred for 1 h, aged at room temperature for 4 h, and then washed repeatedly with deionized water until no sulfate could be detected. The

suspension was then filtrated and dried at 65 °C for 24 h. The dry material was crushed and stored in a desiccator for futher use. The synthesis of Fe–Mn nanowire by using hydrothermal treatment method Basic Solution was prepare from 50ml of 8M NaOH with 1 g of binary oxide prepared, maintaining temprature at 170°C by using an oven controlled temperature and time of reaction was 10h. The precipitates were formed in down of Teflon (nano wire) and then filtered, washed repeatedly with deionized water, and dried at 65°C for 24h.

### 2.3 Characterization. Structure factors, Morphology, composition components

Subjected to X-Ray Diffraction (Cu K $\alpha$  radiation 40 kVand 30 mA) the pattern of particles w recorded. . The microscopic morphologies of the synthesized Fe–Mn particles were observed by using a Scanning electron microscope The chemical state of elements in samples was determined by Energy Dispersive X Ray Spectrometry

Table(1) XRD data

Sample	Fe:Mn (1:1)	Fe:Mn (1:1)N.W.
2 $\theta$ (Deg)	35.518 62.7234	33.963 9 64.4729
4sin $\theta$	1.221 2.081	1.168 2.133
cos $\theta$	0.956 0.855	0.956 0.845
Plane (hkl)	(311) (440)	(311) (400)
Interplaner Spacing d( Å)	2.52542 1.47460	2.63738 1.44409
$\beta$ FWHM (Rad)	0.0741 0.0520	0.0750 0.0433

## 3. RESULTS AND DISCUSSION.

X-ray diffraction pattern of Fe–Mn binary oxide is illustrated in Fig. 1. The pattern with two high broad peaks at 35.4 and 62.52, identified according to d spacing of 0.253 and 0.148 nm, respectively, are eventually of magnetite[12,13]. The patterns of Fe–Mn binary oxides with Fe: Mn molar ratio 1:1 nanowire, is shown in Fig. 2

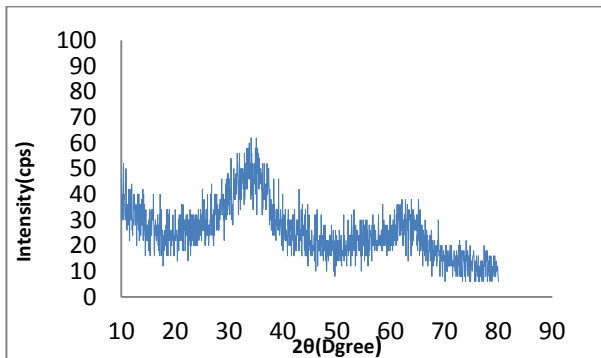


Fig (1) The X-Ray diffraction pattern of sample prepared

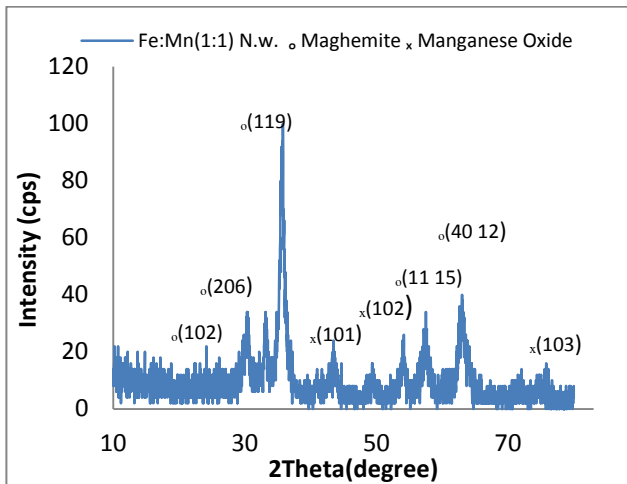


Fig. (2) X-Ray diffraction Fe-Mn (1:1)nanowire

**3.1 Crystallite Particle Size Calculation**

From this study, considering the peak at degrees, average crystallite particle size has been estimated by using Debye-Scherrer formula. Inter-planar spacing between atoms (d-spacing) is calculated

using Bragg's Law and enumerated in Table.1.

$$D = k\lambda / (\beta \cos \theta) \dots\dots\dots(1)$$

$$d_{hkl} = n\lambda / (2 \sin \theta) \dots\dots\dots(2)$$

where,  $\lambda$  is wave length for CuK $\alpha$  radiation (0.1540 nm),  $\beta$  is FWHM (full width at half maximum),  $\theta$  is diffraction angle, d is d-spacing,  $k$  is the geometric factor related to crystallite shape equal to 0.89 and D is crystallite size shown in table 2.

**Table (2) structure factors**

Sample	Fe:Mn (1:1)	Fe:Mn (1:1)N.W.
crystallite size D (nm)	2.29	2.53
strain $\epsilon_s$	0.102	0.1137
Dislocation Density (m <sup>-2</sup> )	1.9*10 <sup>17</sup>	1.56*10 <sup>17</sup>

**3.2. Instrumental Broadening**

An appreciable broadening in diffraction pattern was observed with particle size is less than 100 nm. The broadening is attributed to particle size and strain. The observed line broadening is eventually used to estimate the average size of the particles.

The total broadening of the diffraction peak is, however, due to sample and the instrument. The sample broadening is described by

$$\beta_{hkl} \cos \theta = \left(\frac{k\lambda}{D}\right) + (4\epsilon_s \sin \theta) \dots\dots\dots(3)$$

where D is the average particle size,  $\epsilon_s$  is strain and  $\beta_{hkl}$  is instrumental broadening The total broadening equation is thus described by the instrument-corrected broadening  $\beta_D$  corresponding to the diffraction peak of nanomaterial, estimated, using the relation:

$$\beta_D^2 = [(\beta^2)_{measured} - (\beta^2)_{instrumental}] \dots\dots\dots(4)$$

The size and strain of the experimentally observed broadening of several peaks are computed simultaneously using *least squares method* and presented in Fig.3. When, particle size becomes smaller, due to size effect, the peaks become broad and widths larger. The broadening of peak may also occur due to micro strains of the crystal structure arising from defects like dislocation and twinning [14,15]. NWs possessing a cubic structure tend to coalesce into the bicrystalline or tricrystalline NWs whereas NWs with the hexagonal structure prefer to grow as single-crystalline NWs [16].

Thermal oxidation has received considerable attention because of simplicity technical and large-scale synthesis of high-quality Fe<sub>2</sub>O<sub>3</sub> NWs [17, 18,19,20,21,22,23,24, 25];

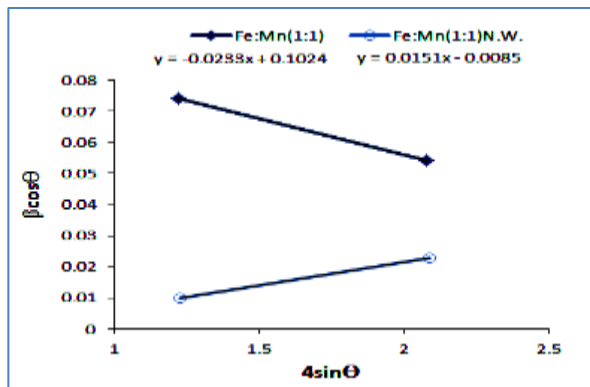


Fig.3. Williamson Hall Plot

**3.3 Strain-induced broadening** arising from crystal imperfections and distortion are related by  $\epsilon_s \sim \beta_s / \tan \theta$ . A remarkable property of Eq. (3) is the dependency on  $\cos \theta$ . The W-H method does not follow a  $1/\cos \theta$  dependency as in the Scherrer equation, but instead varies with  $\tan \theta$ . This fundamental difference allows for a separation of reflection broadening when both microstructural causes - small crystallite size and microstrain - occur together. The different approaches illustrated below assume the size and strain broadening are additive components of the total integral breadth of a Bragg peak [26]. The distinct  $\theta$  dependencies of both effects are assumed to be responsible for the separation of size and strain broadening in the analysis of Williamson and Hall. Addition of the Scherrer equation and  $\epsilon_s \sim \beta_s / \tan \theta$  results in Rearranging Eq.(3) gives the term  $(\beta \cos \theta)$  was plotted with respect to  $(4 \sin \theta)$  for the preferred orientation peaks of nanomaterial. Accordingly, the slope and y-intersect of the fitted line represent strain, for Fe<sub>2</sub>O<sub>3</sub> NWs nucleation during the growth process, we can see shear stress could be produced inside the

Fe<sub>2</sub>O<sub>3</sub> NWs. Fe<sub>2</sub>O<sub>3</sub> NWs become longer, and the stress becomes larger. When the shear stress accumulates to some extent will form in the Fe<sub>2</sub>O<sub>3</sub> NWs. Eventually, the modulated structures accompany with the formation the single-crystalline NWs[16].so the negative strain means uniaxial compressions and positive strain means uniaxial expansion.

**3.4 the dislocation density ( $\delta$ )** which represents the amount of defects in the sample in table 2 is calculated using the relation [27]:

$$\delta = 1/D^2 \dots \dots \dots (5)$$

According to Eq.(5) there is inverse relation between dislocation density and crystallite size so that dislocation density increases when the values of crystallite size decreases, when the dislocation density increases, that means there is strong strain happened with the grain. This behavior is very clear in figures1 and 2 because the peak intensity decreases and the broadening increase with decreasing Fe oxide concentration so nanocrystalline diameter decreases with an increase in crystallite size.

The strain, which is calculated from the slope of (W-H) plot; as well as the dislocation density which is calculated from Eq(5) increases with decreasing the crystallite size.

**3.5 Morphology and composition components**

In order to investigate the morphology of obtained materials by SEM. Fig. 4. shown SEM image of Fe–Mn binary oxides samples , in the picture appears that Nano crystalline particles composed of small particle from which we can know that the diameter size 62.5nm. The control of the monodisperse size is very important because the properties of nano crystal strongly depend upon the dimension of nanoparticles[28]. nanoparticles have spherical shape. In some places, various sizes of the particles (small and large size) are observed, A cluster of agglomerated particles can clearly be seen, so shows the average length~( 1500-2000)nm and the average diameter 63nm of nanowire, the values were calculated according imagej program.

in fig(4,5,6and7). EDS spectrum showed the strong peaks of Fe,Mn and O. The composition components of particulate nanocomposite formed by co-precipitation synthesis Fe, Mn and O. These results demonstrate the purity of the synthesis results and small amount of carbon contamination which is due to the carbon tape).

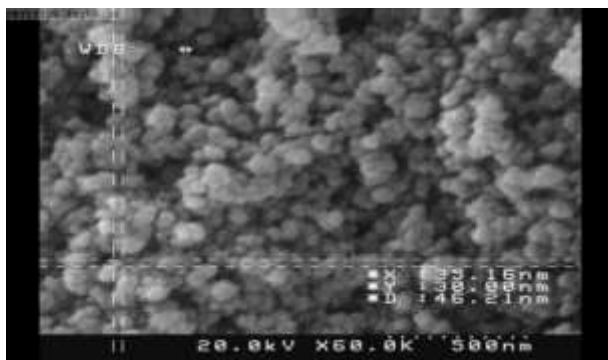
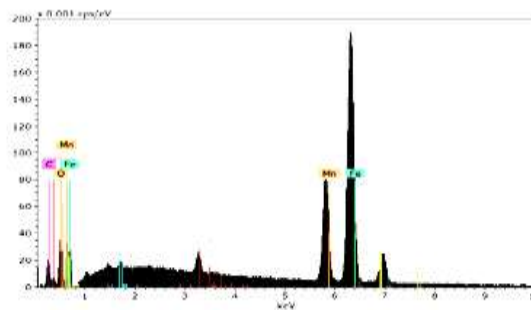


Fig (4) SEM images of of Fe-Mn (1:1)



Fig(5) EDS spectrum for Fe:Mn (1:1)nanoparticles

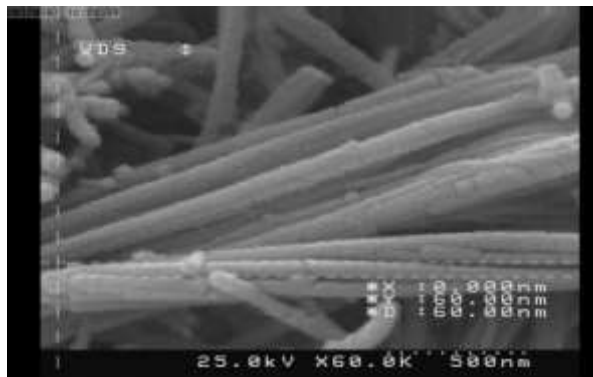
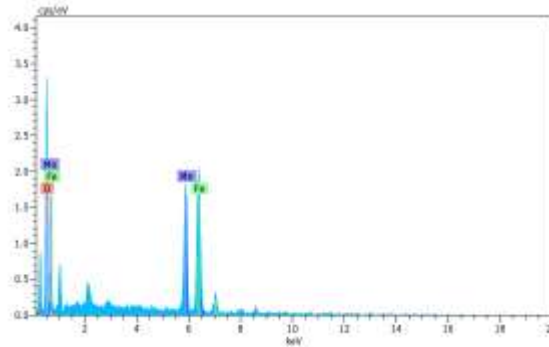


Fig (6) SEM images of of Fe-Mn (1:1) NWs



Fig(7)EDS spectrum for Fe:Mn (1:1) NWs

**4. CONCLUSIONS**

A cheap co-precipitation method has been used to synthesize nanoparticles of Fe–Mn binary oxides. The resulting nanoparticles determines the formation of structure factors whose average size is affected by the size of individual particles. in fact, in bigger particles so the degree of agglomeration related with strain of particle, NWs nucleation during the growth process by hydrothermal treatment of nanoparticles in 170 °C for 10h . SEM show nanoparticles and nano wires and EDS spectrum shown the strong peaks of Fe,Mn and O.

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